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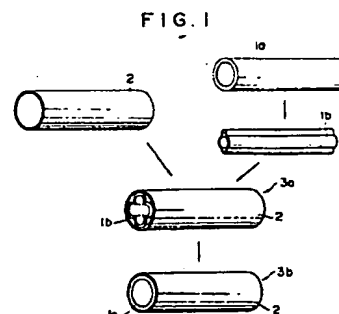
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54 **Shape memory hollow body and method of working the same.**

57 A shape memory hollow body which is molded from a polyurethane elastomer produced by prepolymer process from a raw material composed of a difunctional diisocyanate, difunctional polyol, and difunctional chain extender containing active hydrogen in a molar ratio of 2.00-1.10 : 1.00 : 1.00-0.10, said polyurethane elastomer containing NCO groups and OH groups in almost equal amounts at the terminals of the polymer chains and having a glass transition point in the range of -50 to 60°C and a crystallinity of 3 to 50 wt%, said hollow body remembering its as-molded basic shape but taking on a deformed shape which is given when it is deformed at a temperature higher than the glass transition point of the polymer and lower than the molding temperature and then set when it is cooled in the deformed state to a temperature lower than the glass transition point.

A method of working the above-mentioned shape memory hollow body (1a) remembering its basic shape, said methodprising the steps of de-

forming the hollow body (1a) at a temperature higher than the glass transition point and lower than the molding temperature and then cooling it in the deformed state to a temperature lower than the glass transition point, thereby setting it in the deformed state, combining the deformed hollow body (1b) with another member, (2), and heating the combination (3a) to a temperature higher than the glass transition point, thereby causing it to restore its basic shape and achieving firm bonding of the hollow body (1a) to the member.



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SHAPE MEMORY HOLLOW BODY AND METHOD OF WORKING THE SAME

The present invention relates to a hollow body molded from a thermoplastic polyurethane elastomer having the shape memory function and also to a method of working the same.

Polyurethane having a glass transition point as low as about -40°C exhibits a low modulus even at low temperatures and hence can be used in the application areas for ordinary natural rubber and synthetic rubber. On the other hand, polyurethane having a glass transition point as high as about $100\text{--}110^{\circ}\text{C}$ exhibits a high modulus and good abrasion resistance even at high temperatures and hence finds use as a raw material for artificial wood. ("Glass transition point" will be referred to as T_g hereinafter.)

Meanwhile, the present inventors have proposed a shape memory polyurethane elastomer (in Japanese Patent Laid-open No. 293214:1986). It provides a molding which takes on an as-molded shape and deformed shape when it is heated or cooled to a proper temperature. In other words, the molding takes on a deformed shape when it is deformed at a temperature lower than the molding temperature and then cooled to a temperature below T_g while being kept deformed, so that the deformation is set. The deformed molding restores its original shape when it is heated to a temperature higher than T_g and lower than the molding temperature.

The reference cited above discloses two kinds of shape memory polyurethane elastomers. One is composed of an isophorone-based isocyanate, a polyol, and an adduct of trimethylol propane with tolylene diisocyanate (as a chain extender). The other is composed of 2,4-tolylene diisocyanate, a polyol, and 1,4-butanediol (as a chain extender).

These polyurethane elastomers contain a large amount of excess NCO groups at the terminals of the polymer molecules so that they exhibit rubber elasticity at a temperature higher than the T_g . In addition, the first one has crosslinkage formed by a trifunctional chain extender. Polyurethane elastomers having crosslinkage resemble a thermosetting polymer and hence are limited in the freedom of fabrication. That is, they are very difficult to process by injection molding, extrusion molding, blow molding, or the like.

It is an object of the present invention to provide a hollow body molded from a shape memory polyurethane elastomer which is thermoplastic and capable of melt molding (such as injection molding and extrusion molding) and exhibits rubber elasticity at temperatures higher than its T_g approximate to room temperature. It is another object of the present invention to provide a method of work-

ing said hollow body.

The first aspect of the present invention resides in a shape memory hollow body which is molded from a polyurethane elastomer produced by prepolymer process from a raw material composed of a difunctional diisocyanate, difunctional polyol, and difunctional chain extender containing active hydrogen in a molar ratio of $2.00\text{--}1.10 : 1.00 : 1.00\text{--}0.10$, said polyurethane elastomer containing NCO groups and OH groups in almost equal amounts at the terminals of the polymer chains and having a glass transition point in the range of -50 to 60°C and a crystallinity of 3 to 50 wt%, said hollow body remembering its as-molded basic shape but taking on a deformed shape which is given when it is deformed at a temperature higher than the glass transition point of the polymer and lower than the molding temperature and then set when it is cooled in the deformed state to a temperature lower than the glass transition point.

The second aspect of the present invention resides in a method of working the above-mentioned shape memory hollow body remembering its basic shape, said method comprising the steps of deforming the hollow body at a temperature higher than the glass transition point and lower than the molding temperature and then cooling it in the deformed state to a temperature lower than the glass transition point, thereby setting it in the deformed state, combining the deformed hollow body with another member, and heating the combination to a temperature higher than the glass transition point, thereby causing it to restore its basic shape and achieving firm bonding of the hollow body to the member.

Fig. 1 is a schematic drawing showing the steps of fitting an inner pipe of shape memory polyurethane elastomer into a pipe.

Fig. 2 is a sectional view showing the joining of a shape memory hollow body to a pipe.

The conventional shape memory polyurethane elastomer molding depends on rigid allophanate linkages for its rubber elasticity at a temperature higher than its T_g . The allophanate linkages result from the intermolecular crosslinking which takes place when excess terminal NCO groups react with urethane linkages. By contrast, the shape memory polyurethane elastomer in the present invention is a thermoplastic chain polymer which is composed of a difunctional isocyanate, polyol, and chain extender according to a specific formulation but contains no excess terminal NCO groups. It is characterized by a specific crystallinity, a T_g approximate to room temperature, and a specific ratio of moduli measured at temperatures above and below

the Tg. This polyurethane elastomer has the partial crystallization instead of the intermolecular crosslinking. Therefore, it is a thermoplastic polymer of chain structure; nevertheless, it exhibits rubber elasticity at a temperature higher than its Tg and takes on the as-molded shape and deformed shape at temperatures above and below its Tg. In other words, it has the shape memory function. This thermoplastic polyurethane elastomer can be easily molded into a hollow body by melt molding such as injection molding and extrusion molding.

The thermoplastic polyurethane elastomer in the present invention should preferably have a crystallinity in the range of 3 to 50 wt%. With a crystallinity lower than 3 wt%, the polymer will have a low rubber elasticity at a temperature higher than its Tg. Conversely, with a crystallinity higher than 50 wt%, the polymer will have a high rubber elasticity at a temperature higher than its Tg, with the result that the ratio of moduli at temperatures 10°C above and below its Tg is smaller.

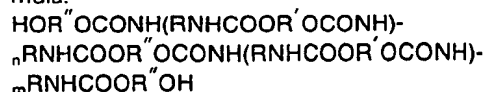
The polyurethane polymer in the present invention is produced from the following raw materials, which are illustrative only and not limitative.

The first raw material is a difunctional isocyanate which is represented by the general formula OCN-R-NCO , where R is a group having no or one or two benzene rings. It includes, for example, 2,4-toluene diisocyanate, 4,4'-diphenylmethane diisocyanate, carbodiimide-modified 4,4'-diphenylmethane diisocyanate, and hexamethylene diisocyanate.

The second raw material is a difunctional polyol which is represented by the general formula OH-R'-OH , where R' is a group having no or one or two benzene rings. The second raw material may also be a reaction product of said difunctional polyol and a difunctional carboxylic acid or cyclic ether. It includes, for example, polypropylene glycol, 1,4-butane glycol adipate, polytetramethylene glycol, polyethylene glycol, and an adduct of bisphenol-A with propylene oxide.

The third raw material is a difunctional chain extender containing active hydrogen which is represented by the general formula OH-R''-OH , where R'' is a $(\text{CH}_2)_n$ group or a group having no or one or two benzene rings. It includes, for example, ethylene glycol, 1,4-butane glycol, bis(2-hydroxyethyl)hydroquinone, an adduct of bisphenol-A with ethylene oxide, and an adduct of bisphenol-A with propylene oxide.

The thus produced polyurethane elastomer may be represented by the following general formula.



where m is 1-16 and n is 0-16.

The following illustrates the production of the shape memory polyurethane elastomer. First, a prepolymer is prepared by reacting, in the absence of catalyst, an isocyanate component and a polyol component in the ratio shown in Table 1. To the prepolymer is added a chain extender in the ratio shown in Table 1. The resulting mixture is heat-cured to give the shape memory polyurethane elastomer, which has the basic physical properties as shown in Table 1.

In Table 1, Tg represents the glass transition point (°C) and E/E' represents the ratio of the tensile modulus at a temperature 10°C lower than the Tg to the tensile modulus at a temperature 10°C higher than the Tg. The crystallinity (wt%) was measured by X-ray diffractometry.

The hollow body pertaining to the present invention is made of the above-mentioned shape memory polyurethane elastomer. It remembers its basic shape when it is molded. It is given its second shape (deformed shape) when it is deformed at a temperature higher than its Tg and lower than its molding temperature and then cooled in the deformed state to a temperature lower than its Tg. Subsequently, the deformed hollow body is combined with another member, and the combination is heated to a temperature higher than its Tg, so that the hollow body restores its basic shape, firmly joining itself to the member.

Table 1

| Raw materials and molar ratio | | M.W. | 1 | 2 | 3 | 4 | 5 | 6 | 7 | 8 | 9 | 10 |
|--|---|---------------------|------|------|------|------|------|------|------|------|------|------|
| Diisocyanate | 2,4-toluene diisocyanate | 174 | 1.5 | | | 1.5 | | | | | | |
| | 4,4'-diphenylmethane diisocyanate | 250 | | | | | 1.5 | | | 1.5 | 1.5 | 1.5 |
| | 4,4'-diphenylmethane diisocyanate (carbodiimide-modified) | 290 | | | | | | 1.5 | | | | |
| | ditto | 303 | | 1.5 | 1.5 | | | | | | | |
| Polyol | hexamethylene diisocyanate | 168 | | | | | | | 1.5 | | | |
| | polypropylene glycol | 400 | | | | | | | | | | |
| | ditto | 700 | | | 1.0 | 1.0 | 1.0 | 1.0 | 1.0 | 1.0 | 1.0 | 1.0 |
| | ditto | 1000 | | 0.88 | | | | | | | | |
| | 1,4-butanediol adipate | 600 | | | | | | | | | | |
| | ditto | 1000 | | | | | | | | | | |
| | ditto | 2000 | | | | | | | | | | |
| | polytetramethylene glycol | 650 | | | | | | | | | | |
| | ditto | 850 | | | | | | | | | | |
| | ditto | 1000 | | | | | | | | | | |
| Chain extender | polyethylene glycol | 600 | | | | | | | | | | |
| | bisphenol-A + propylene oxide | 800 | 1.0 | | | | | | | | | |
| | ethylene glycol | 62 | | | | | | | | 0.51 | | |
| | 1,4-butanediol | 90 | 0.51 | | | | | | | | 0.51 | |
| | bis(2-hydroxyethyl)hydroquinone | 198 | | | | | | | | | | |
| | bisphenol-A + ethylene oxide | 327 | | | | | | | | | | |
| | ditto | 360 | | 0.51 | 0.51 | 0.51 | 0.51 | 0.51 | 0.51 | | | |
| | bisphenol-A + propylene oxide | 360 | | | | | | | | | | 0.51 |
| Measured values of physical properties | | T _g (°C) | 24 | -10 | 15 | -11 | 14 | 16 | -45 | 9 | 6 | 12 |
| | | E/C | 170 | 73 | 69 | 23 | 129 | 133 | 20 | 117 | 128 | 97 |
| | | Crystallinity (wt%) | | 20 | 20 | 30 | | | 25 | | | |

Table 1 (continued)

| Raw materials and molar ratio | | M.W. | 11 | 12 | 13 | 14 | 15 | 16 | 17 | 18 | 19 | 20 |
|--|---|---------------------|------|------|-----|------|------|------|------|------|------|------|
| Diisocyanate | 2,4-toluene diisocyanate | 174 | | | | | | | | | | |
| | 4,4'-diphenylmethane diisocyanate | 250 | 1.5 | 1.5 | 1.5 | 1.2 | 1.8 | 1.35 | 1.35 | 1.35 | 1.35 | 1.35 |
| | 4,4'-diphenylmethane diisocyanate (carbodiimide-modified) | 290 | | | | | | | | | | |
| | ditto | 303 | | | | | | | | | | |
| | hexamethylene diisocyanate | 168 | | | | | | | | | | |
| Polyol | polypropylene glycol | 400 | | | | | | | | | | |
| | ditto | 700 | 1.0 | 1.0 | | 1.0 | 1.0 | 1.0 | | | | |
| | ditto | 1000 | | | | | | | 1.0 | | | |
| | 1,4-butaneglycol adipate | 600 | | | | | | | | 1.0 | | |
| | ditto | 1000 | | | | | | | | | 1.0 | |
| | ditto | 2000 | | | | | | | | | | 1.0 |
| | polytetramethylene glycol | 650 | | | | | | | | | | |
| | ditto | 850 | | | | | | | | | | |
| | ditto | 1000 | | | | | | | | | | |
| | polyethylene glycol | 600 | | | 1.0 | | | | | | | |
| Chain extender | bisphenol A + propylene oxide | 800 | | | | | | | | | | |
| | ethylene glycol | 62 | | | | | | | | | | |
| | 1,4-butane glycol | 90 | | | | | | | | | | |
| | bis(2-hydroxyethyl)hydroquinone | 198 | | 0.51 | | | | | | | | |
| | bisphenol A + ethylene oxide | 327 | 0.51 | | | 0.21 | 0.81 | 0.36 | 0.36 | 0.36 | 0.36 | 0.36 |
| Measured values of physical properties | ditto | 360 | | | | | | | | | | |
| | bisphenol A + propylene oxide | 360 | | | | | | | | | | |
| | | | | | | | | | | | | |
| | | Tg (°C) | 16 | -7 | -6 | -4 | 25 | 5 | -22 | 10 | -18 | -45 |
| | | E/E | 111 | 49 | 12 | 105 | 5.2 | 37 | 81 | 100 | 29 | 30 |
| | | Crystallinity (wt%) | | 20 | 30 | | 20 | 25 | | | 25 | 25 |

Table 1 (continued)

| Raw materials and molar ratio | | M.W. | 21 | 22 | 23 | 24 | 25 | 26 | 27 | 28 | 29 | 30 |
|--|---|---------------------|------|------|------|------|------|------|-----|------|------|------|
| Diisocyanate | 2,4-toluene diisocyanate | 174 | | | | | | | | | | |
| | 4,4'-diphenylmethane diisocyanate | 250 | 1.35 | 1.35 | 1.35 | 1.5 | 1.5 | 1.35 | 1.5 | 1.4 | 1.3 | 1.2 |
| | 4,4'-diphenylmethane diisocyanate (carbodiimide-modified) | 290 | | | | | | | | | | |
| | ditto | 303 | | | | | | | | | | |
| Polyol | hexamethylene diisocyanate | 168 | | | | | | | | | | |
| | poly(propylene glycol) | 400 | | | | | | 1.0 | | | | |
| | ditto | 700 | | | | | | | | | | |
| | ditto | 1000 | | | | | | | | | | |
| | 1,4-butanediol adipate | 600 | | | | | | | | | | |
| | ditto | 1000 | | | | | | | | | | |
| | ditto | 2000 | | | | | | | | | | |
| | poly(tetramethylene glycol) | 650 | 1.0 | | | | | | | | | |
| | ditto | 850 | | 1.0 | | | | | | | | |
| | ditto | 1000 | | | 1.0 | | | | | | | |
| Chain extender | poly(ethylene glycol) | 600 | | | | | | | | | | |
| | bisphenol-A + propylene oxide | 800 | | | | | | | | | | |
| | ethylene glycol | 62 | | | | | | | | | | |
| | 1,4-butanediol | 90 | | | | | | | | | | |
| | bis(2-hydroxyethyl)hydroquinone | 198 | | | | | | | | 0.51 | 0.41 | 0.31 |
| | bisphenol-A + ethylene oxide | 327 | 0.36 | 0.36 | 0.36 | 0.43 | 0.35 | 0.36 | | | | |
| | ditto | 360 | | | | | | | | | | |
| | bisphenol-A + propylene oxide | 360 | | | | | | | | | | |
| | | | | | | | | | | | | |
| | | | | | | | | | | | | |
| Measured values of physical properties | | | | | | | | | | | | |
| | | T _g (°C) | -18 | -30 | -38 | 5 | 8 | 23 | 26 | 21 | 19 | 19 |
| | | E/E' | 33 | 18 | 40 | 33 | 100 | 126 | 140 | 125 | 108 | 101 |
| | | Crystallinity (wt%) | 25 | 25 | | 25 | 15 | 15 | 10 | 15 | 15 | 15 |

Table 1 (continued)

| Raw materials and molar ratio | | M.W. | 31 | 32 | 33 | 34 | 35 | 36 | 37 | 38 | 39 | 40 |
|--|--|---------------------|------|------|------|------|------|------|------|------|------|------|
| Diisocyanate | 2,4-toluene diisocyanate | 174 | | | 1.5 | | | | | | | |
| | 4,4'-diphenylmethane diisocyanate | 250 | 1.59 | 1.68 | | 1.3 | 1.7 | 1.59 | 1.68 | 1.5 | 1.5 | 1.81 |
| | 4,4'-diphenylmethane diisocyanate (carbodiimide-modified) | 290 | | | | | | | | | | |
| | ditto | 303 | | | | | | | | | | |
| Polyol | hexamethylene diisocyanate | 168 | | | | | | | | | | |
| | polypropylene glycol | 400 | | | | | | | | | | |
| | ditto | 700 | 1.0 | 1.0 | | 1.0 | 1.0 | 1.0 | 1.0 | 1.0 | | |
| | ditto | 1000 | | | | | | | | | | |
| | 1,4-butanediol adipate | 600 | | | | | | | | | | |
| | ditto | 1000 | | | | | | | | | | |
| | ditto | 2000 | | | | | | | | | | |
| | polytetramethylene glycol | 650 | | | | | | | | | | |
| | ditto | 850 | | | | | | | | | | |
| | ditto | 1000 | | | | | | | | | | |
| Chain extender | polyethylene glycol | 600 | | | | | | | | | | |
| | bisphenol-A + propylene oxide | 800 | | | 1.0 | | | | | 1.0 | 1.0 | 1.0 |
| | ethylene glycol | 62 | | | | 0.31 | 0.71 | 0.51 | 0.51 | | | |
| | 1,4-butane glycol | 90 | | | | | | | | 0.51 | | |
| | bis(2-hydroxyethyl)hydroquinone | 198 | | | 0.51 | | | | | | 0.51 | 0.81 |
| Measured values of physical properties | bisphenol-A + ethylene oxide | 327 | | | | | | | | | | |
| | ditto | 360 | | | | | | | | | | |
| | bisphenol-A + propylene oxide | 360 | 0.51 | 0.51 | | | | | | | | |
| | ditto | 360 | | | | | | | | | | |
| Measured values of physical properties | | T _g (°C) | | 10 | 11 | 22 | 2 | 15 | 11 | 12 | 35 | 40 |
| | | E/E | | 126 | 126 | 107 | 83 | 122 | 100 | 135 | 124 | 138 |
| | | Crystallinity (wt%) | | 15 | 20 | 15 | 20 | 15 | 15 | 10 | 10 | 5 |
| | | | | | | | | | | | | 48 |
| | | | | | | | | | | | | 152 |

EXAMPLES

The invention will be described in more detail with reference to the following examples, which are not intended to restrict the scope of the invention.

Example 1

In this example, a hollow body (inner pipe) made of the shape memory polyurethane elastomer was fitted into an outer pipe according to the following procedure.

First, a polymer having a T_g of 48°C was prepared by prepolymer process according to the formulation of sample No. 40 shown in Table 1. The polymer was made into a cylindrical hollow body **1a** having an outside diameter of 10.4 cm and a wall thickness of 5 mm by extrusion molding.

This cylindrical hollow body **1a** was heated to about 65°C and then collapsed by pressing. It was cooled to about 40°C in the collapsed state so that it was set deformed. Thus there was obtained a collapsed hollow body **1b**.

The collapsed hollow body **1b** was inserted into a steel pipe **2** having an inside diameter of 10 cm. (The resulting combination is indicated by **3a**.)

The collapsed hollow body **1b** inserted into the steel pipe **3a** was heated by blowing hot air (about 65°C), so that it restored its original cylindrical shape. Thus there was obtained a composite pipe **3b** composed of a steel pipe **2** and a cylindrical hollow body **1a** of polyurethane elastomer which was firmly bonded to the inside of the steel pipe.

This example shows that the use of the shape memory function overcomes the difficulty which was encountered in the past when the inside of a small steel pipe was coated with a polyurethane elastomer.

Example 2

In this example, a pipe made of the shape memory polyurethane elastomer was joined to another pipe as shown in Fig. 2.

A polymer having a T_g of 40°C was prepared by prepolymer process according to the formulation of sample No. 39 shown in Table 1. The polymer was molded into a straight pipe (not shown) having an outside diameter of 5 cm and a wall thickness of 3 mm. This straight pipe remembers its as-molded shape. The end **4** (3 cm long) of the pipe was heated to about 60°C and then contracted until the outside diameter was 4 cm.

The pipe was cooled to about 35°C , with the end contracted, so that the contracted shape was set. Thus there was obtained a pipe **5** as shown in Fig. 2.

The end **4** of the pipe **5** was inserted into a pipe **6** having an inside diameter of 4.5 cm. When they are heated to about 60°C , the end of the pipe **5** restored its original shape, joining itself firmly to the inside of the pipe **6**. The above-mentioned joining procedure may be applied to joining an elbow made of shape memory polymer to a steel pipe.

As the above-mentioned examples show, the thermoplastic polyurethane elastomer having the shape memory function can be molded into a hollow body of any shape by melt molding such as injection molding, extrusion molding, and blowing molding.

In addition, the shape memory hollow body of the present invention remembers its basic shape, takes on its second shape (e.g., reduced or enlarged diameter, and bending), and restores its basic shape after it has been combined with another member. This property makes it possible to join the hollow body firmly to another member.

Claims

1. A shape memory hollow body which is molded from a polyurethane elastomer produced by prepolymer process from a raw material composed of a difunctional diisocyanate, difunctional polyol, and difunctional chain extender containing active hydrogen in a molar ratio of 2.00-1.10 : 1.00 : 1.00-0.10, said 1:polyurethane elastomer containing NCO groups and OH groups in almost equal amounts at the terminals of the polymer chains and having a glass transition point in the range of -50 to 60°C and a crystallinity of 3 to 50 wt%, said hollow body remembering its as-molded basic shape but taking on a deformed shape which is given when it is deformed at a temperature higher than the glass transition point of the polymer and lower than the molding temperature and then set when it is cooled in the deformed state to a temperature lower than the glass transition point.

2. The shape memory hollow body as claimed in Claim 1, wherein the difunctional diisocyanate is one which is represented by the general formula OCN-R-NCO , where R is a group having no or one or two benzene rings.

3. The shape memory hollow body as claimed in Claim 1, wherein the difunctional polyol is one which is represented by the general formula OH-R'-OH , where R' is a group having no or one or two benzene rings.

4. The shape memory hollow body as claimed in Claim 1, wherein the difunctional chain extender containing active hydrogen which is represented by the general formula $\text{OH}-\text{R}''-\text{OH}$, where R'' is a $(\text{CH}_2)_n$ group or a group having no or one or two benzene rings. 5

5. The shape memory hollow body as claimed in Claim 1, wherein the polyurethane elastomer polymerized by prepolymer process is one which is represented by the general formula below: 10
 $\text{HOR}''\text{OCONH}(\text{RNHCOOR}'\text{OCONH})_n\text{RNHCOOR}''\text{OCONH}(\text{RNHCOOR}'\text{OCONH})_m\text{RNHCOOR}''\text{OH}$
 where m is 1-16 and n is 0-16.

6. A method of working the shape memory hollow body remembering its basic shape as claimed in Claim 1, said method comprising the steps of deforming the hollow body at a temperature higher than the glass transition point and lower than the molding temperature and then cooling it in the deformed state to a temperature lower than the glass transition point, thereby setting it in the deformed state, combining the deformed hollow body with another member, and heating the combination to a temperature higher than the glass transition point, thereby causing it to restore its basic shape and achieving firm bonding of the hollow body to the member. 15 20 25

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FIG. 1

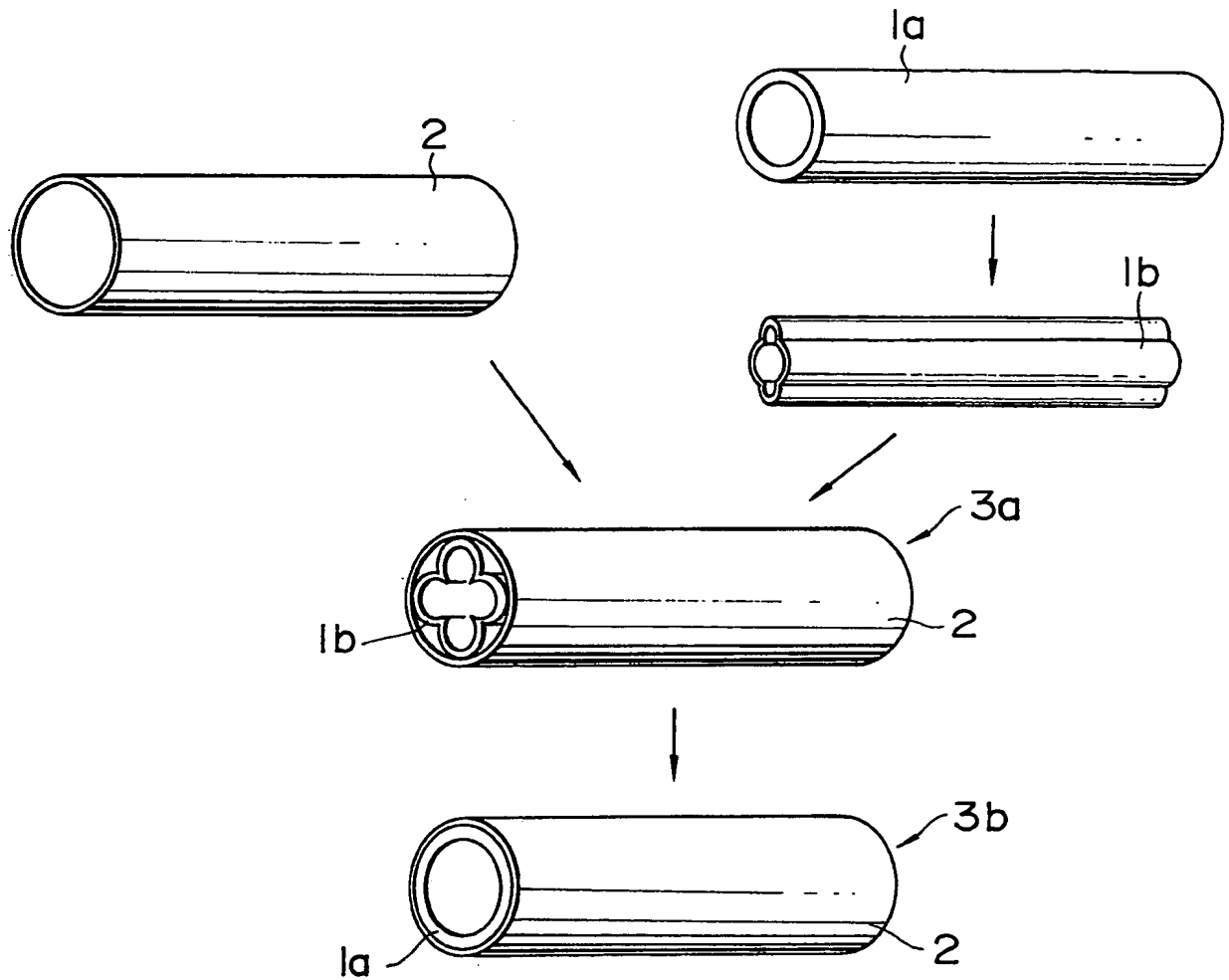
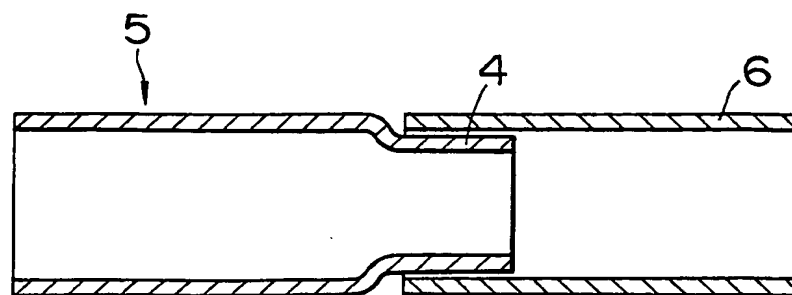


FIG. 2



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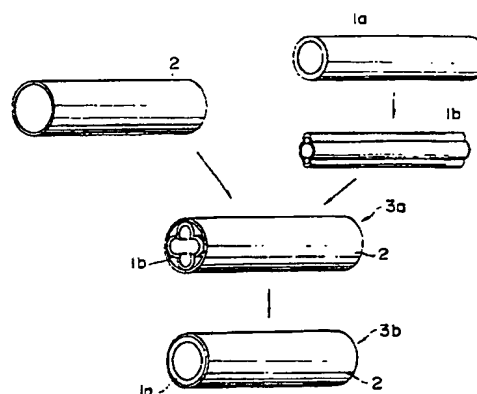
Shape memory hollow body and method of working the same.

A shape memory hollow body which is molded from a polyurethane elastomer produced by prepolymer process from a raw material composed of a difunctional diisocyanate, difunctional polyol, and difunctional chain extender containing active hydrogen in a molar ratio of 2.00-1.10 : 1.00 : 1.00-0.10, said polyurethane elastomer containing NCO groups and OH groups in almost equal amounts at the terminals of the polymer chains and having a glass transition point in the range of -50 to 60 °C and a crystallinity of 3 to 50 wt%, said hollow body remembering its as molded basic shape but taking on a deformed shape which is given when it is deformed at a temperature higher than the glass transition point of the polymer and lower than the molding temperature and then set when it is cooled in the deformed state to a temperature lower than the glass transition point.

A method of working the above-mentioned shape memory hollow body (1a) remembering its basic shape, said methodprising the steps of deforming the hollow body (1a) at a temperature higher than the glass transition point and lower than the molding temperature and then cooling it in the de-

formed state to a temperature lower than the glass transition point, thereby setting it in the deformed state, combining the deformed hollow body (1b) with another member, (2), and heating the combination (3a) to a temperature higher than the glass transition point, thereby causing it to restore its basic shape and achieving firm bonding of the hollow body (1a) to the member.

FIG. 1





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Application Number

EP 89 11 9214

| DOCUMENTS CONSIDERED TO BE RELEVANT | | | |
|---|--|---|---|
| Category | Citation of document with indication, where appropriate, of relevant passages | Relevant to claim | CLASSIFICATION OF THE APPLICATION (Int. Cl.5) |
| A | US-A-3 624 045 (STIVERS) * Column 2, line 62 - column 4, line 75; claim 1 * | 1-6 | B 29 C 71/00 C 08 G 18/10 C 08 G 18/32 |
| A | FR-A-1 549 412 (RAYCHEM) * Page 2, right-hand column, line 8 - page 4, left-hand column, line 37; claims 1-4 * | 1-6 | |
| A | FR-A-1 534 052 (DUNLOP) | 1-6 | |
| A | PATENT ABSTRACTS OF JAPAN, vol. 11, no. 163 (C-424)[2610], 26th May 1987; & JP-A-61 293 214 (MITSUBISHI) * Abstract * | 1-6 | |
| A | DE-A-1 704 160 (GUTHOFFNUNGSHUTTE) | 1-6 | |
| | | | TECHNICAL FIELDS SEARCHED (Int. Cl.5) |
| | | | B 29 C C 08 G |
| The present search report has been drawn up for all claims | | | |
| Place of search THE HAGUE | | Date of completion of the search 21-08-1990 | Examiner BOURGONJE A. F. |
| CATEGORY OF CITED DOCUMENTS | | | |
| X : particularly relevant if taken alone Y : particularly relevant if combined with another document of the same category A : technological background O : non-written disclosure P : intermediate document | | T : theory or principle underlying the invention E : earlier patent document, but published on, or after the filing date D : document cited in the application L : document cited for other reasons & : member of the same patent family, corresponding document | |